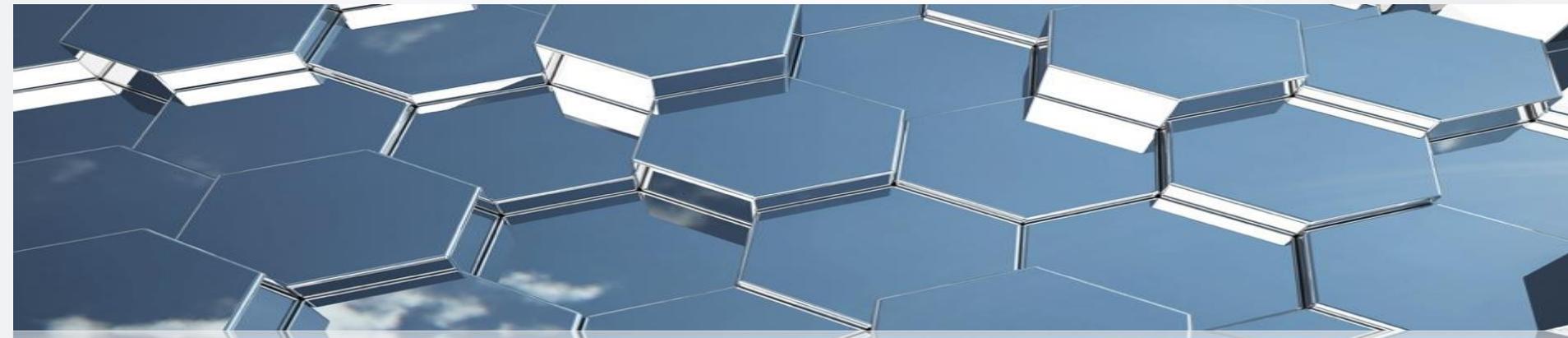


New Insights into Controlling Sub-micron Failure Mechanisms in Composites using Discrete Functionalized Multiwall Carbon Nanotubes, Molecular Rebar™

C.P. Bosnyak



Special Thanks

Organizing Committee for the Symposium celebrating the 75th Year of Prof. Alexander Chudnovsky.

**In particular : Victor Berdichevsky
Ephraim Suhir**

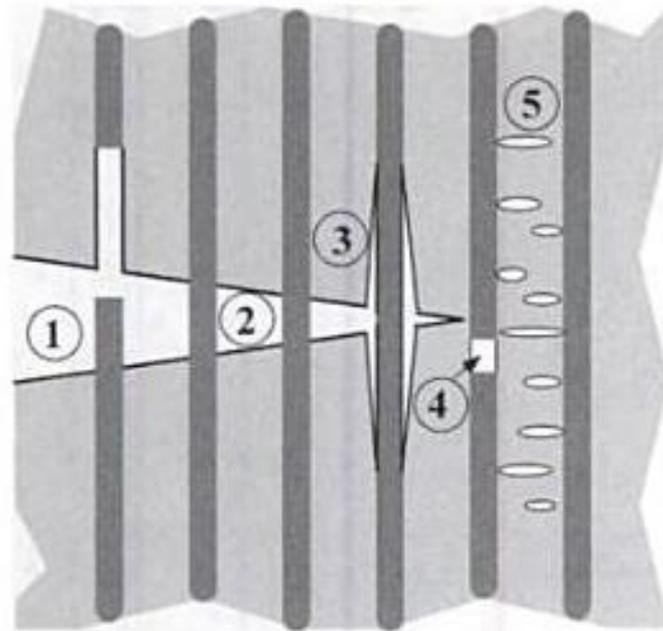
Unmet Needs in the Composites Industry

- Mass Transportation – train, bus
 - Light weighting – fuel economy
 - Surface quality (out of the mold)
 - Flame retardant – all mass transport
- Wind
 - Increase turbine capacity using longer blades
 - Improved resistance to failure (especially seawater)
- Aerospace
 - Reliability
 - Lighter and larger parts at same stiffness
 - Corrosion resistance
 - Better static electricity management (lightning strikes)
 - Repair



Fiberglass → FG & carbon fiber hybrids
→ FG, CF, Nanotechnology hybrids

Basic failure mechanisms of epoxy-fiber composites



(a) In-plane damage.

1. Fiber Pull-Out.
2. Fiber Bridging.
3. Fiber/Matrix Debonding
4. Fiber Failure.
5. Matrix Cracking.

Initial scale of failure in epoxies on submicron scale – **aim of nanocomposites is to control failure at this scale**

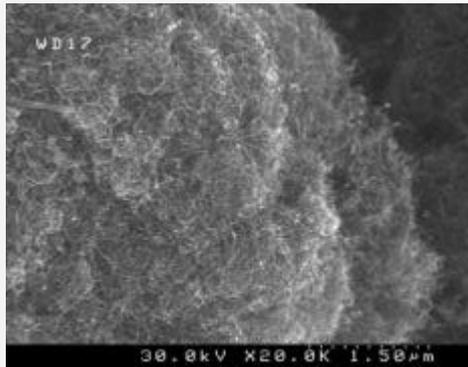
Failure in composites often due to interlaminar cracking

Literature

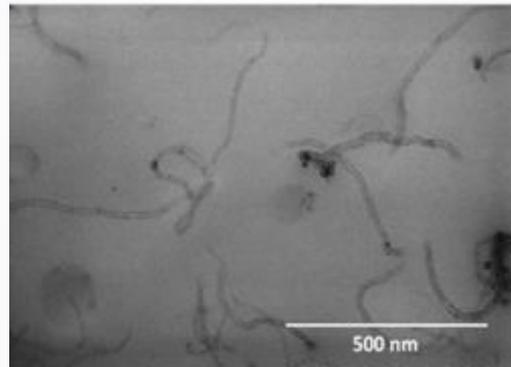
- Several studies on epoxy and carbon nanotubes since 1993.
- Almost all studies show MWNT agglomerates.
- Tubes typically contain 10-20% residual catalysts and soot.
- Mechanical properties show bonded tubes much better than non-bonded.
- Conductivity highly dependent on amount and dispersion state of tubes.
- For improved dispersions need to modify the thermodynamics of the tube surface and detangle
 - Highly dependent on epoxy-hardener type

Molecular Rebar™ (MR)

Key Competitive Advantage



Clumps of entangled CNT's



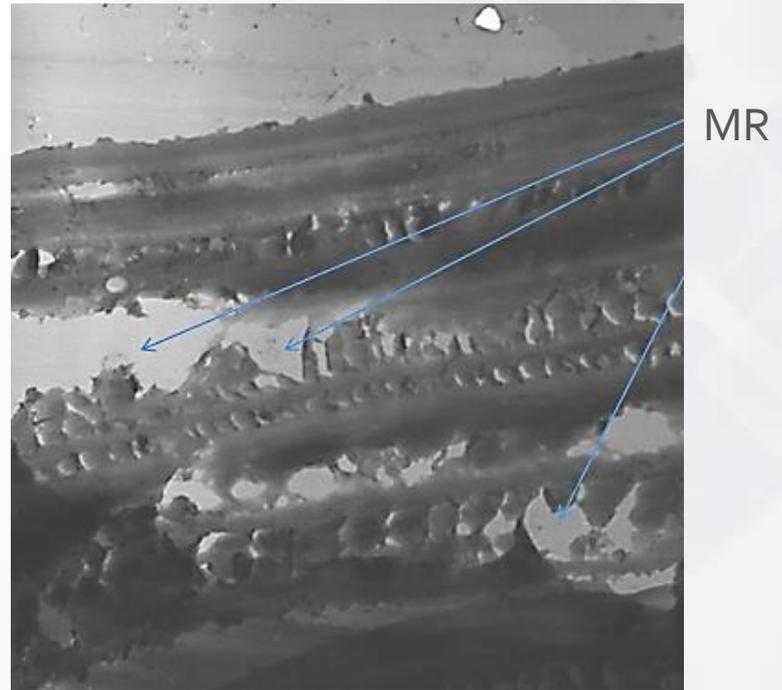
Clean, Discrete Tubes
10 walls
Aspect Ratio 60-100
13nm x 800nm
Surface modified for Dispersion & Bonding



Integration into parts

Advantages of MR

- Discrete, well-dispersed carbon nanotubes can move within plies without filtering
- Bonded to matrix to improve inter-laminar shear modulus and strength
 - Reduced mechanical stress gradients
 - Reduced thermal coefficient of expansion – reduced stress gradients (also cure)
 - Increased crack resistance



 Increased durability, improved safety

Challenges for Nanotube Reinforcing Thermoset Composites

- Many varied performance requirements for a broad set of applications
- Many varieties of chemistries
- Impossible to test all variations...
- *so - Can we begin to develop a set of “Materials Design Rules for Durability” with MR?*

Chudnovsky Approach to Durability Assessment

- Consider the environment the material is to survive in
 - Stress state, Temperature, Time, Surrounding medium, etc.
- Quantify the initial material structure
 - Consider Homogeneity vs Heterogeneity relative to scale of measurement
- Identify & quantify the possible mechanisms of failure dependent on the external imposed variables
- Model – “Predict with Understanding”
 - Challenge is to predict Failure **Initiation** as well as Failure Propagation

Notch Sensitivity Index, K

X. Nui et al. "Notch Sensitivity of Pipe Grade Polyethylene and Polybutylene"
ANTEC 2000 - Proceedings of 58th Annual Technical Conference & Exhibition,
Vol. XLVI, Orlando Fl., May 7-11, Society of Plastics Engineers, p 3162-3106

Test Unnotched versus Notched

Impact Energy

$$K_e = \lim_{a \rightarrow 0} \frac{W}{W(a)} = 1 + \frac{W_i}{W_p}$$

Time to creep failure

$$K_t = \lim_{a \rightarrow 0} \frac{t}{t(a)} = 1 + \frac{t_i}{t_p}$$

Which part of this ratio do nanotubes influence?

Materials that have higher notch sensitivity are probably less reliable under load considering the presence of defects.

Thermoset Resin Systems

- Bisphenol A / Jeffamine D230
 - Cure profile: 2hr 100°C, 1hr 150°C; Tg 95°C
- Bisphenol F / TETA
 - Cure profile: 2 hr 100°C; Tg 105°C
- TGDDM / DDS
 - Cure profile: 2 hr 120°C, 1hr 150°C, 2hr 177°C, 4 hr 200°C; Tg 290°C

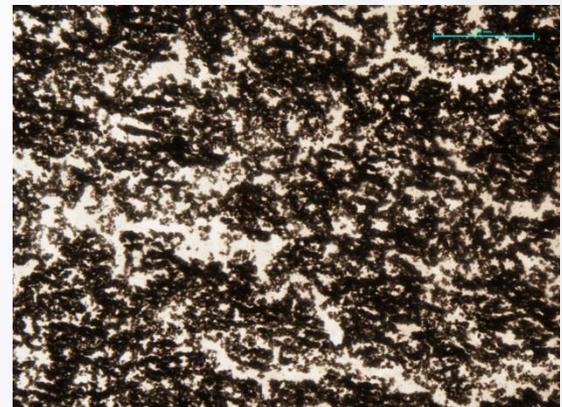
Challenge of nanotube dispersion

- High surface area of MR ~ 250 m²/g means small changes in surface energy differences strongly affect degree of dispersion.
- Challenge is to keep the dispersion while the epoxy components are mixed and during curing.

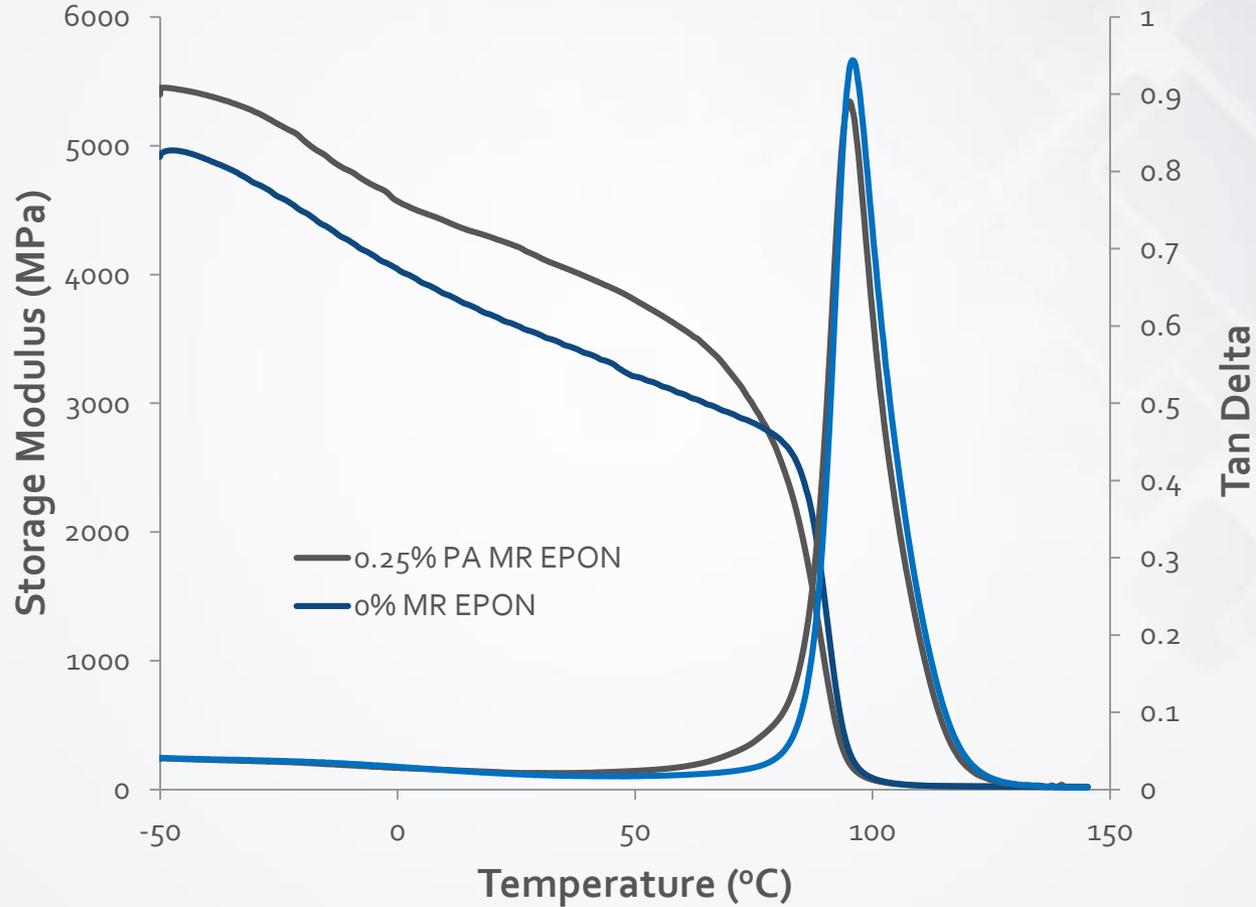
0.25% w/w MR treated



0.25% w/w MR not treated



Thermomechanical Characterization of Epon 828 + MR



No significant sub T_g transitions to -50°C

Testing Procedures

Single edge notched bending

- SENB - ASTM E1820, Test conditions: -78 to 60°C, 10mm/min.

- $$K_{1c} = \frac{PS}{BW^{\frac{3}{2}}} f(x) \quad s = \text{span (4")}, P = \text{load}$$

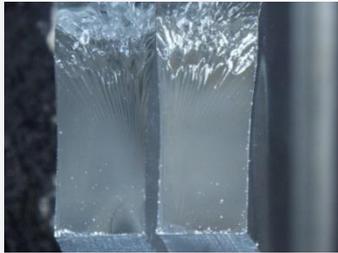
$$\text{Where } f(x) = 3x^{\frac{1}{2}} \left[\frac{1.99 - x(1-x)(2.15 - 3.93x + 2.7x^2)}{2(1+2x)(1-x)^{\frac{3}{2}}} \right] \quad x = a/w$$

- Notched with a band-saw and crack initiated with a razor blade
- If crack moves faster to part failure than strain rate in SENB then $W_i > W_p$
- Single edge notched tensile specimen for Bisphenol F epoxy
 - 1mm notch depth
 - $\sigma_{\min}/\sigma_{\max}$ 0.1
 - σ_{\max} 16-17MPa
 - 1Hz, 25°C
 - Ramp test 0.25mm/min

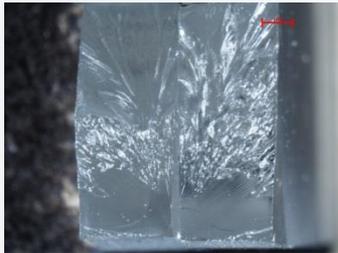
Unnotched Flexure w/wo MR

Control (Blue line)

60°C



25°C



0°C

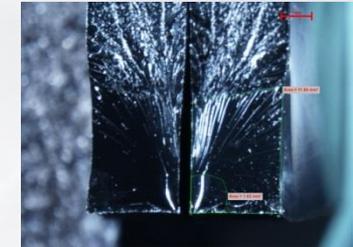


-78°C

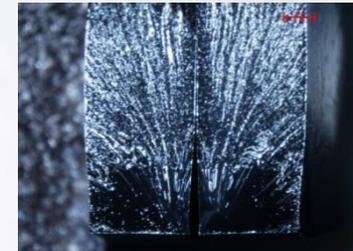


0.25% MR type 1 (Black line)

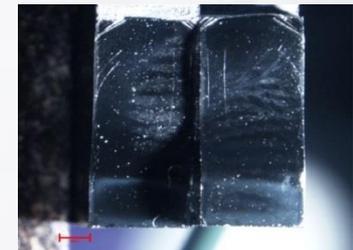
60°C



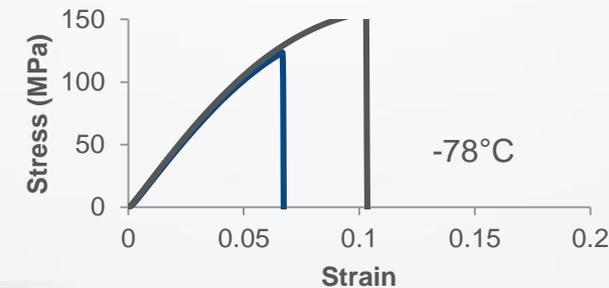
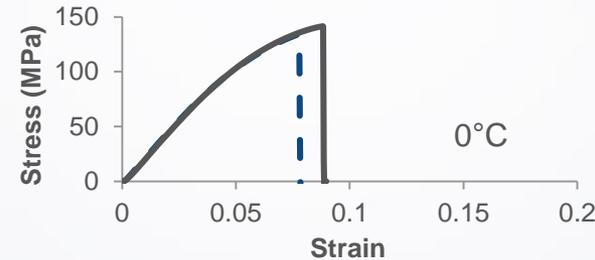
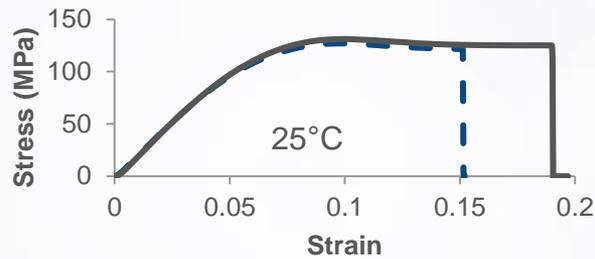
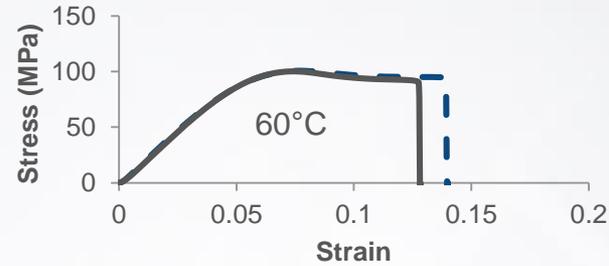
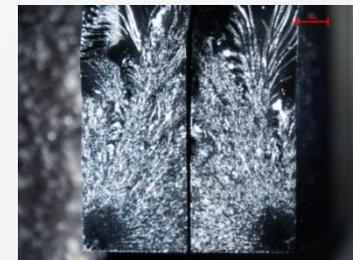
25°C



0°C



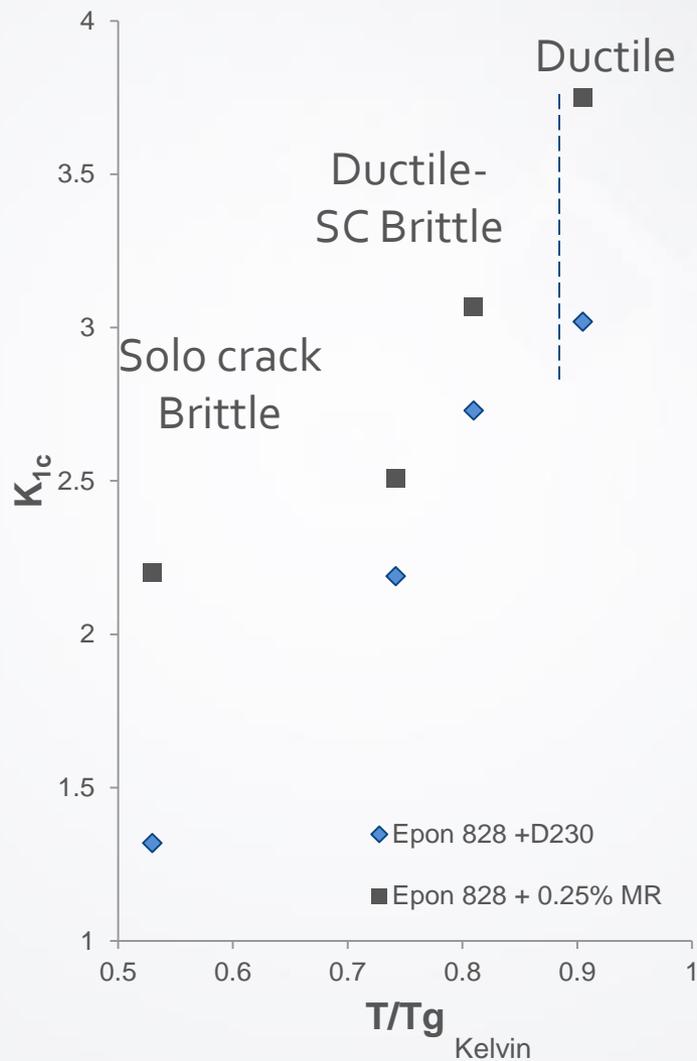
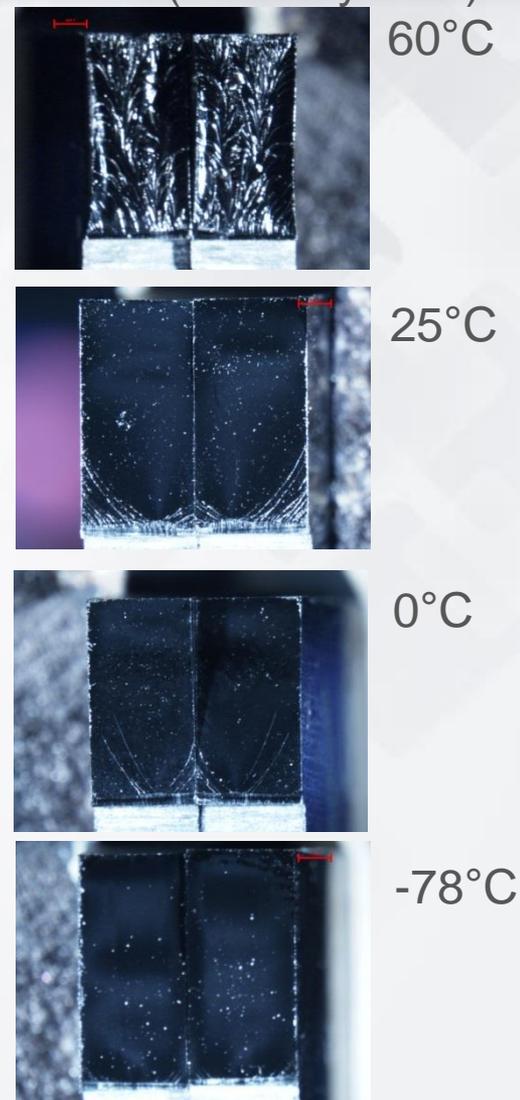
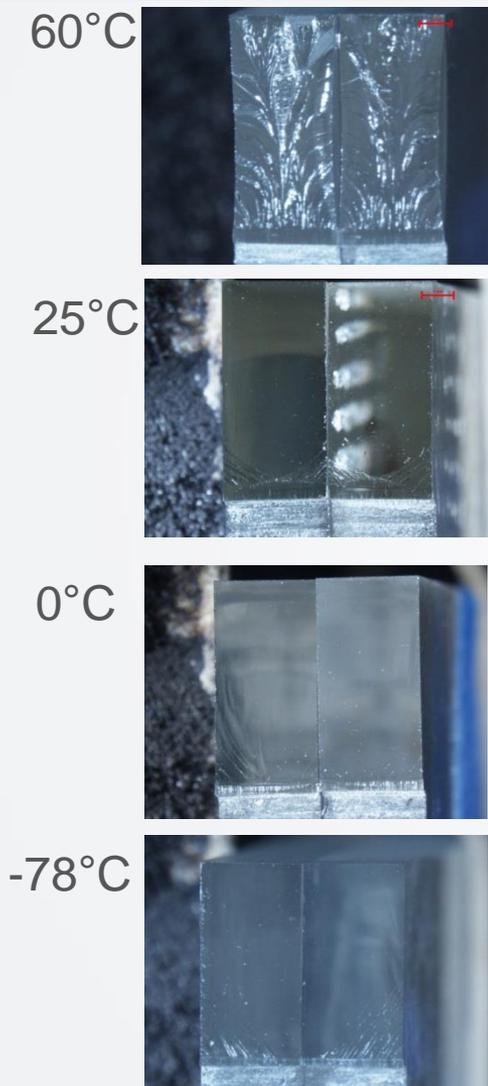
-78°C



SENB, $a/w \sim 0.5$; w/wo MR

Control (Blue symbol)

0.25% MR (Black symbol)



Notch Sensitivity Index values

Bisphenol A epoxy with Jeffamine

Epoxy + %MR	Temp °C	K_{1c}	NSI
0	-78	1.3	11
0	0	2.2	19.2
0	25	2.73	18
0.25%	-78	2.2	4.6
0.25%	0	2.5	14.2
0.25%	25	3.1	9.6

The values of K_{1c} and NSI, taken together with observations of the fractured surfaces are consistent with both W_i as well as W_p increasing with MR addition, but W_p is increasing relatively more than W_i on MR addition based on reduced NSI values.

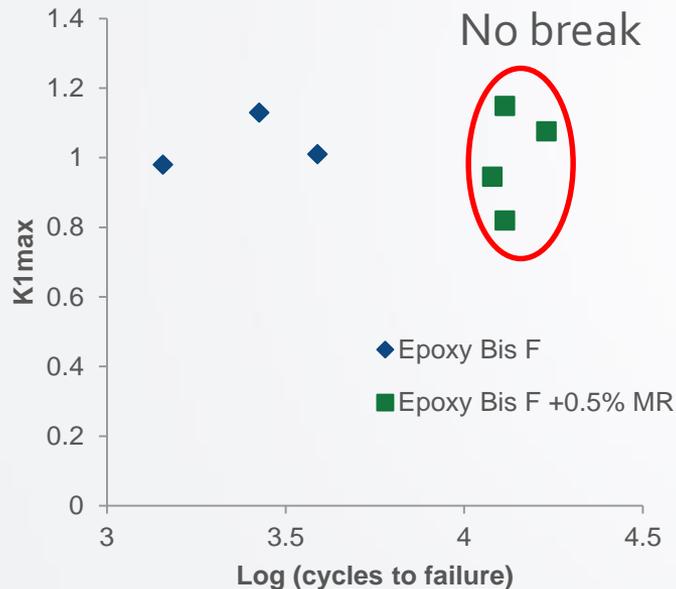
Bisphenol F Epoxy cured with TETA

T_g = 105 °C; MR fully exfoliated in Bis-F Epoxy

0% wt MR K_{1c} (25°C) = 2.3 MPa.m^{0.5}

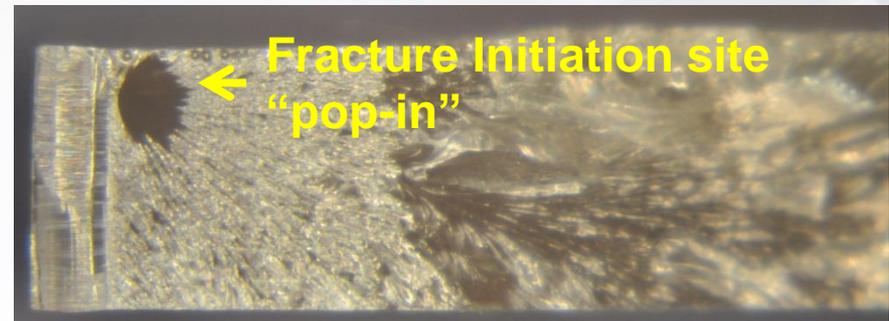
0.25% wt MR K_{1c} (25°C) = 2.7 MPa.m^{0.5}

0.5% wt MR K_{1c} (25°C) = 3.3 MPa.m^{0.5}

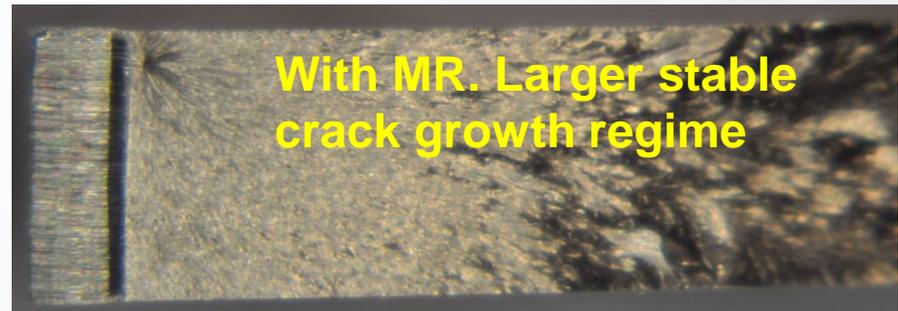


Ramp test

Fracture Surface of Bis F Epoxy



Bis F Epoxy and 0.5% wt MR

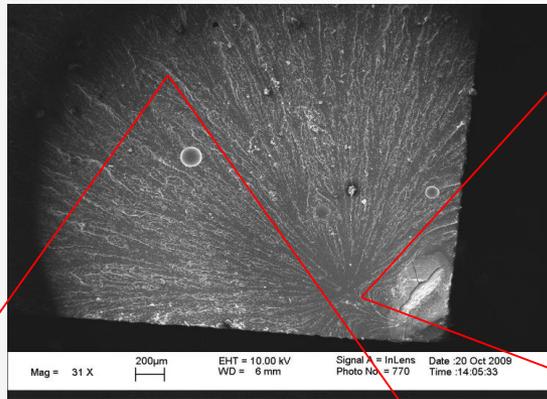


- Increased fatigue fracture resistance at least 5X

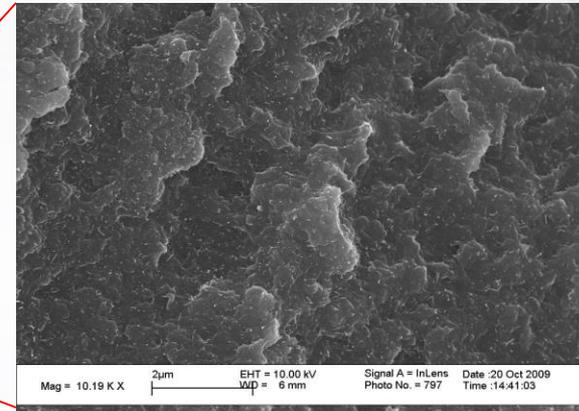
Electron Microscopy Gold sputtered

Fracture surface of unnotched tensile bar of 1% wt MR Bisphenol F epoxy

31X

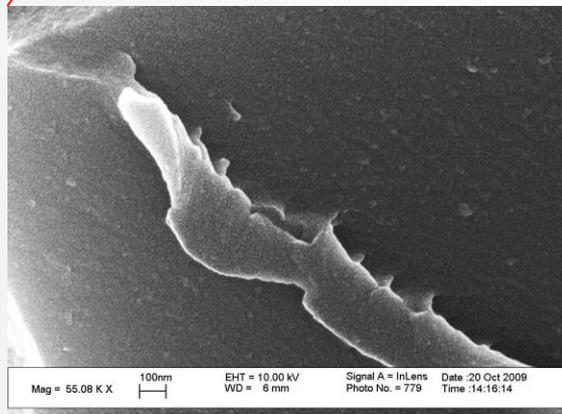


10KX



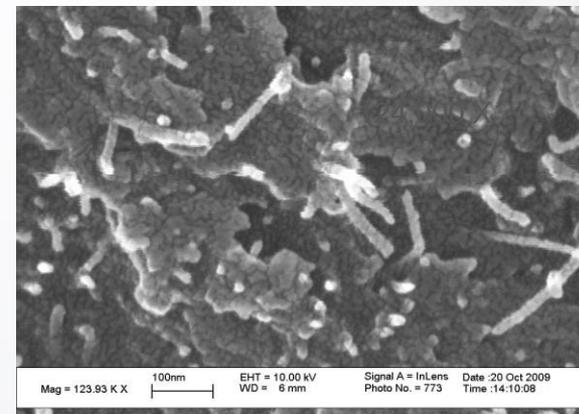
Fast Fracture Zone

55KX



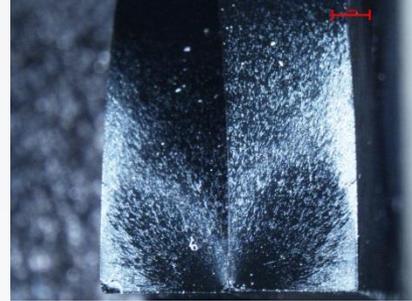
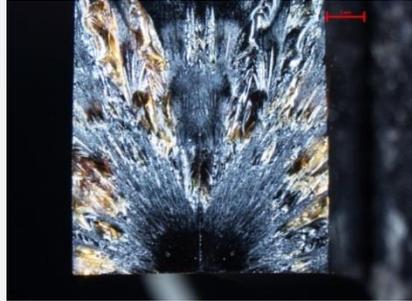
Crack Initiation Zone

124KX



MY 721-DDS Tetra glycidyl, High Tg (290°C) Aircraft composites

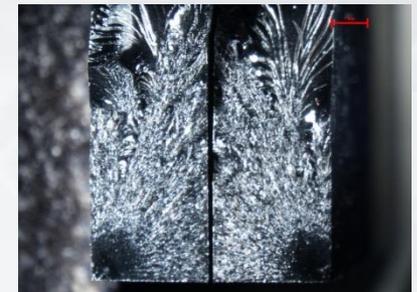
MY 721 134°C



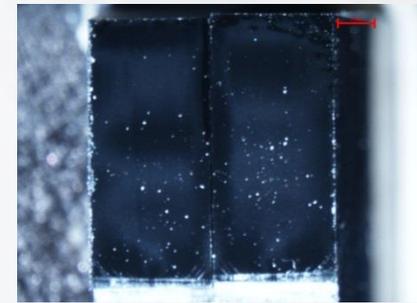
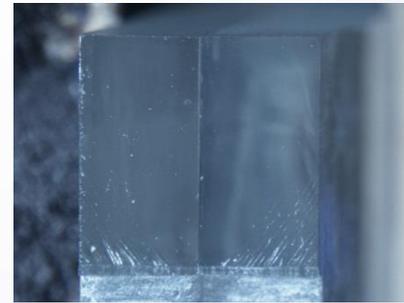
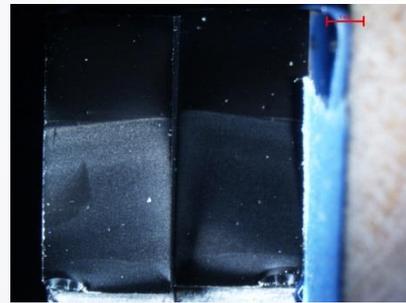
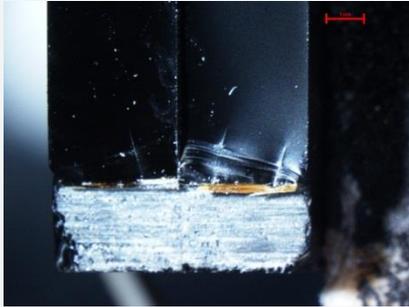
unnotched



Epon 828 -78°C



notched



- Difficult to get NSI and K_{1c} values \Rightarrow MY721 has very low W_i and low W_p even at 134°C.
- MR improves fracture resistance as seen by fracture surfaces. Should translate to improved fatigue or creep failure resistance.
- Scaling of fracture mechanism does not scale simply to T/T_g . Must also include understanding of molecular crosslink structure.

Conclusions

- Addition of MR to epoxy provides a microcrack bridging mechanism that:
 - Improves resistance to crack initiation and has greater crack dissipation energy, particularly for slow crack growth.
 - Smaller “pop-in” zone even though higher potential energy
 - Observed many more sub-micron scale fracture events (deltoids)
 - When a crack is moving fast the localized energy at the crack tip cleaves the carbon nanotubes at the crack surface)
 - Provides less sensitivity to material defects, lower NSI values
 - **Greatly improved material fatigue lifetime (and presumably creep fracture resistance) as well as impact toughness.**

Conclusions, cont.

- The higher the epoxy T_g and greater the crosslink density the more sensitive the material is to fracture via defects.
 - Nanotube dispersion quality has to be excellent.
- Can place MR (discrete, bonded tubes) between fiber plies to better manage interlaminar fatigue or creep failure events.
- Demonstrated a systematic approach to understanding failure initiation and propagation events provides much insight into proper design and expectations of performance for carbon nanotubes in epoxy systems.

Acknowledgements

People

A. Chudnovsky for over 33 years of wisdom and guidance in the selection and design of materials for durability

S. Peddini, University of Texas for imaging SEM

M. Gauthier, MRD for some sample preparation

Materials

Bisphenol F from the Dow Chemical Company

Bisphenol A from 3M Corporation

MY721 from Huntsman and recommendations for curing.